

Mechanism of the order–disorder phase transition, and glassy behavior in the metal-organic framework $[(\text{CH}_3)_2\text{NH}_2]\text{Zn}(\text{HCOO})_3$

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Transitions associated with orientational order–disorder phenomena are found in a wide range of materials and may have a significant impact on their properties. In this work, specific heat and ^1H NMR measurements have been used to study the phase transition in the metal-organic framework (MOF) compound $[(\text{CH}_3)_2\text{NH}_2]\text{Zn}(\text{HCOO})_3$. This compound, which possesses a perovskite-type architecture, undergoes a remarkable order–disorder phase transition at 156 K. The $(\text{CH}_3)_2\text{NH}_2^+$ (DMA⁺) cationic moieties that are bound by hydrogen bonds to the oxygens of the formate groups (N–H...O ~ 2.9 Å) are essentially trapped inside the basic perovskite cage architecture. Above 156 K, it is the orientations of these moieties that are responsible for the disorder, as each can take up three different orientations with equal probability. Below 156 K, the DMA⁺ is ordered within one of these sites, although the moiety still retains a considerable state of motion. Below 40 K, the rotational motions of the methyl groups start to freeze. As the temperature is increased from 4 K in the NMR measurements, different relaxation pathways can be observed in the temperature range approximately 65–150 K, as a result of a “memory effect.” This dynamic behavior is characteristic of a glass in which multiple states possess similar energies, found here for a MOF. This conclusion is strongly supported by the specific heat data.

multiferroic | organic-inorganic solids

Metal-organic framework (MOF) compounds are extended organic–inorganic crystalline solids in well-defined geometric structures (1, 2). Because of the exciting possibility of dynamic and related behavior with advanced device applications, considerable effort is currently being devoted to creating previously undescribed MOF compounds (3, 4). We recently reported a series of multifunctional MOF compounds that simultaneously exhibit ferromagnetic and ferroelectric behavior associated with orientational ordering (5, 6). These compounds possess the perovskite structure, ABX_3 , with A as the dimethylammonium cation $(\text{CH}_3)_2\text{NH}_2^+$, DMA⁺; B as a divalent transition metal ion (Zn^{2+} , Mn^{2+} , Ni^{2+} , Fe^{2+} , etc.), and X as the formate ion HCO_2^- (5, 6). Because many of the inorganic perovskites are ferroelectrics, it was interesting to find these analogous MOF compounds to also exhibit ferroelectric behavior, but the microscopic mechanism of the onset of ferroelectricity and the concomitant solid/solid-phase transition was not fully understood (5, 6). A clue to the mechanism is provided by the molecular structure. X-ray studies show that $[(\text{CH}_3)_2\text{NH}_2]\text{Zn}(\text{HCOO})_3$ (henceforth DMAZF) has the basic perovskite architecture in which the metal atoms are located at the corners of a cube and are connected via the oxygen atoms of formate linkers (Zn–O–CH–O–Zn) (Fig. 1) (6, 7). The DMA⁺ cation is located in the cavity and is hydrogen-bonded to the formate frame via its amino H atoms to the formate O atoms, and—in analogy with the antiferroelectric phase transition in $\text{NH}_4\text{H}_2\text{PO}_4$ —the making and breaking of the N–H...O bonds in the unit cell

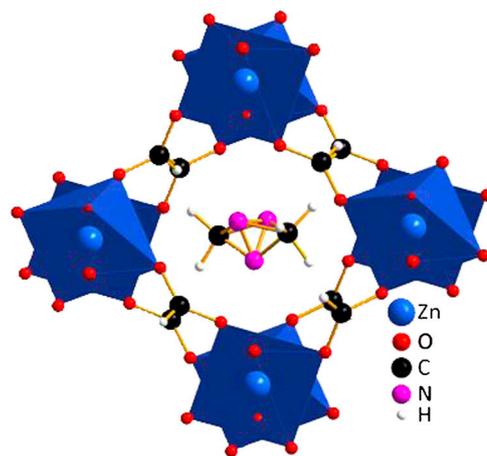


Fig. 1. Crystal structure of $[(\text{CH}_3)_2\text{NH}_2]\text{Zn}(\text{HCOO})_3$ at room temperature. The purple spheres represent the three dynamically disordered sites for the N atom of the DMA⁺ cation.

are the cause of the phase transition and the dipole moment dynamics (8).

Because of the intrinsic weakness of the H bonds, the nitrogen moves between three distinct but equivalent positions at high temperatures ($T > 156$ K), due to thermally induced hindered motion. Upon cooling, the compound undergoes a phase transition as the three sites become inequivalent for a given cavity as far as the nitrogen is concerned, but not enough to freeze the motion. This conclusion is supported by a recent Raman study by Sanchez-Andujar et al. (9) on the manganese analogue of DMAZF showing that as the temperature is lowered, the NH_2 Raman peaks exhibit splittings at 100 K. Moreover, single crystal X-ray determination of the low temperature phase by these authors (9) showed that it belonged to a ferroelectric space group, a result consistent with our dielectric measurements (6). However, no measurements at temperatures close to T_c were reported (9).

In this paper, we report the results of detailed NMR and specific heat (C_p) measurements on DMAZF that show (i) a slowing down of the hopping motion of the caged dimethylammonium moiety, with a minimum in the ^1H spin-lattice relaxation time (T_1) around the phase transition; (ii) an unusual memory effect in the temperature dependence of T_1 at temperatures

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below the phase transition, implying glass-like behavior; and (iii) a peak in C_p/T^3 vs. T data, which corroborates the presence of a glassy phase. To the best of our knowledge, glassy behavior of a single crystalline MOF has not been reported before.

To study the motion of the DMA⁺ ion in relation to the phase transition, we measured the nuclear spin-lattice relaxation time of the ¹H nuclei across the critical temperature $T_c \approx 156$ K, and over a relatively wide temperature range (4–250 K). The T_1 was used as a probe of the local order and molecular dynamics (time scale approximately 10^{-10} – 10^{-12} s) of the DMA⁺ unit and was measured using a spectrometer available at the National High Magnetic Field Laboratory. The standard $\pi/2 - \tau - \pi/2$ saturation recovery procedure was used with a field of 6.267 T (corresponding to 266.8 MHz for protons). The observed ¹H signal was a single, broad peak, containing contributions from all the protons. The sample consisted of 20–30 submillimeter-sized single crystals, synthesized as described elsewhere (6).

Fig. 2 summarizes the behavior of the proton spin-lattice relaxation rate, T_1^{-1} , over a wide temperature range around the transition temperature, T_c . The graph has been divided into three regions (the regions depicted by Fig. 2 A–C), guided by natural breaks in the slope. In the region depicted by Fig. 2C, DMAZF is in the high-temperature paraelectric phase. On lowering the temperature, T_1^{-1} increases steadily and reaches a maximum around T_c , whereafter it starts to decrease as DMAZF enters the ferroelectrically ordered phase (the regions depicted in Fig. 2 A and B).

Our most significant observation from the measurements is that T_1^{-1} is multivalued in the region depicted in Fig. 2B. As long as the sample has not been previously cooled below 40 K, the behavior of the spin-lattice relaxation time is completely reversible over the temperature range 65–250 K in the regions depicted in Fig. 2 B and C. This is the main pathway, as denoted in Fig. 2. However, this reversibility does not hold for the sample if it is cooled below 40 K, and in such cases, T_1^{-1} follows a different path on warming. This unusual result is discussed in detail later in the paper. In the region depicted in Fig. 2A, T_1^{-1} follows a different curve, implying a change in the spin-lattice relaxation pathway.

Spin-lattice relaxation of protons in DMA⁺ has been investigated earlier (10–12), and it is known to be caused by the mod-

ulation of different proton–proton dipolar interactions. These interactions are due to various dynamic processes of the DMA⁺ group, including (10, 11) (i) isotropic and anisotropic tumbling of the DMA⁺ group, (ii) n-fold reorientations of the DMA⁺ group around its diad axis, (iii) internal rotations of the methyl groups around their threefold symmetry axes, and (iv) reorientations of the DMA⁺ cation due to the nitrogen hopping among its three distinct positions.

We can ignore motions (i) and (ii) as they do not preserve the symmetry of DMA⁺ (10, 11) and are therefore expected to be quite hindered, taking place only at very high temperatures, higher than the range in the current study. Therefore, in the following analysis, we consider only the two processes (iii) and (iv), keeping in mind that motion (iv) not only involves the NH₂ motion but also the motion of the CH₃ as a group. Motion (iii) is expected to dominate the relaxation only at lower temperatures, well below the phase transition. Thus, the most dominant motion for the ¹H T_1 at temperatures around the phase transition is motion (iv). It should, however, be stressed that, because only a single relaxation time was observed, it is assumed that also spin diffusion among the various types of protons exists that actually mixes all the relaxation pathways. Therefore, the single relaxation time that is observed is a weighted average value of all relaxations.

The well-rounded maximum in T_1^{-1} (or minimum in T_1 ; see Fig. 3) in the close vicinity of the phase transition suggests that the hopping motion of the DMA⁺ slows down smoothly at T_c , rather than showing a sudden divergence as might be expected from a complete freezing-in of a motional degree of freedom at T_c . Thus, a considerable amount of motion is retained by the DMA⁺ on cooling down through the phase transition. This observation is consistent with the calculation of transition entropy using the C_p curve (vide infra).

The temperature dependence of the relaxation rate that involves the paraelectric phase (the region depicted in Fig. 2C), the region close to the phase transition, and the main path in the ferroelectric phase (the region depicted in Fig. 2B) was analyzed using the Bloembergen–Purcell–Pound formalism (13), in which the relaxation rate is related to the correlation time, τ_c , which is the characteristic time between significant fluctuations in the local magnetic field experienced by a spin due to molecular motions or reorientations of a molecule. As the relaxation rate along this path is dominated by only the two dynamic processes (iii) and (iv), we can write a simplified expression of the total relaxation rate as a sum of two relaxation rates:

$$\begin{aligned} \frac{1}{T_1} &= \frac{1}{T_1^{(1)}} + \frac{1}{T_1^{(2)}} \\ &= C_1 \left(\frac{\tau_{c1}}{1 + (\omega_0 \tau_{c1})^2} + \frac{4\tau_{c1}}{1 + (2\omega_0 \tau_{c1})^2} \right) \\ &\quad + C_2 \left(\frac{\tau_{c2}}{1 + (\omega_0 \tau_{c2})^2} + \frac{4\tau_{c2}}{1 + (2\omega_0 \tau_{c2})^2} \right), \end{aligned} \quad [1]$$

where ω_0 is the ¹H resonance frequency, and C_1 and C_2 are dipolar constants containing proton–proton distances. The correlation times τ_{c1} and τ_{c2} are assumed to obey the Arrhenius law

$$\tau_{ci} = \tau_{0i} \exp(E_{ai}/RT), \quad i = 1, 2, \quad [2]$$

where τ_{0i} is the single particle correlation times and E_{ai} is the activation energies for the two processes. The C_i , τ_{0i} , and E_{ai} are treated as fit parameters.

Fig. 3 shows the inverse temperature dependence of the spin-lattice relaxation time for the aforementioned regions, along with the fit to Eqs. 1 and 2. The yielded parameters are summarized in Table 1.

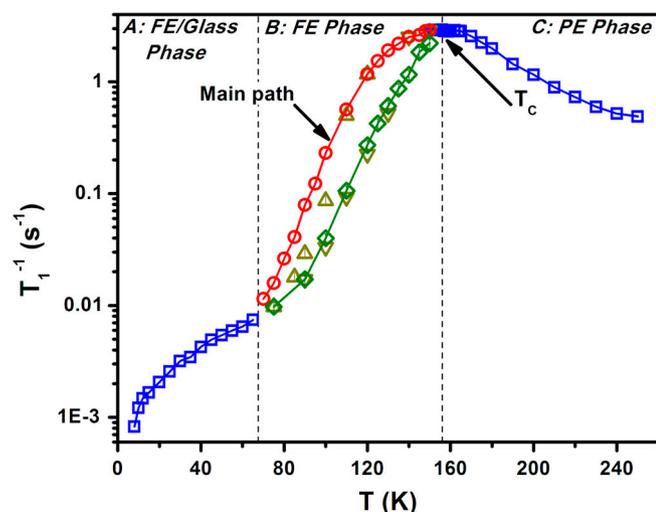


Fig. 2. Temperature dependence of the proton spin-lattice relaxation rate T_1^{-1} of DMAZF. A, B, and C refer to the glass, ferroelectric (FE), and paraelectric (PE) phases, respectively. The main path (circles in the region depicted in B) denotes the reversible temperature dependence behavior as long as the sample is not cooled below approximately 40 K. Refer to the main text for further explanation of the multivalued region depicted in B. Solid lines are guides to the eye, error bars are within symbols, and dashed lines emphasize the division of the relaxation rate in the three regions.

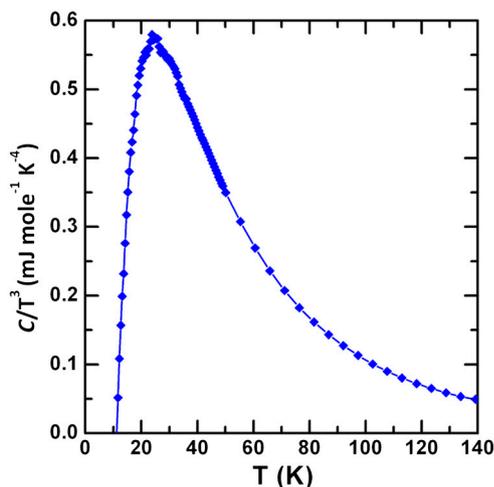


Fig. 5. Temperature dependence of low-temperature specific heat (plotted as C/T^3 vs. T) of $[(\text{CH}_3)_2\text{NH}_2]\text{Zn}(\text{HCOO})_3$. A clear peak around 25 K is a clear manifestation of a glassy phase (see text).

path shown in Fig. 4), occurred only when the sample was cooled relatively quickly (>5 K/min) as compared to the cooling processes of the other, metastable, paths. Analyzing this relaxation pathway in a similar fashion to the main pathway, we obtain the parameters $E_a = 12.2 \pm 0.4$ kJ/mol and $\tau_0 = 4.9 \times 10^{-14}$ s. These values yield the correlation time $\tau_c \approx (5.8 \pm 2.3) \times 10^{-10}$ s at the phase transition. For this relaxation pathway, both the activation energy and the correlation time are higher, compared to the main pathway. Because T_1 depends on the local geometry, these data show that the DMA⁺ moiety finds distinctly different local environments depending on the sample cooling history, again a clear signature of a glassy phase.

The spin-lattice relaxation rate, T_1^{-1} , decreases rapidly from 156 to 65 K, but the rate of decrease slows down significantly as the temperature approaches approximately 40 K. Such behavior is indicative of the opening up of a new relaxation pathway. This is because the dynamic process at these temperatures is dominated by the slowing of the methyl-group internal rotations below approximately 40 K. A similar observation has been reported by Clough and Hill (15) in methyl malonic acid and related compounds.

The T_1^{-1} curve in Fig. 2 can be decomposed into three components: one following the main relaxation pathway, a second showing a maximum around 40 K, and a third that remains essentially constant below about 10 K. The latter two pathways are analogous to the relaxation processes found in many methyl-containing solids.

The existence of a glassy state is further confirmed by our specific heat (C_p) measurements. These measurements indicated that, at low temperatures, C_p was unusually large, an observation that is a clear indication of local ordering (14). Furthermore, for

glasses, the phonon heat capacity at relatively low temperatures in the range of 2–30 K is generally larger than that predicted by the Debye T^3 law (16). The excess specific heat, which is indicated by a peak in the C/T^3 vs. T curve, is generally ascribed to localized vibrations, domain wall motions of the glassy mosaic structure, and/or transverse phonon modes (17, 18). These modes are usually seen in Raman scattering as low-frequency peaks. Because these low frequencies are observed in glasses in which all types of bonding might be involved (metallic, ionic, covalent, etc.), they are recognized as fundamental features of glass dynamics (19). A number of more traditional crystalline materials also exhibit similar low temperature peaks in C/T^3 plots. Such plots have been used to explore the nature of vibrations and disorders in materials such as quartz SiO_2 , and the pyrochlore $\text{Bi}_2\text{Ti}_2\text{O}_7$ (16). In the present case, the C/T^3 plot for the DMAZF also revealed a well-defined broad peak at 25 K (Fig. 5), well below its ferroelectric transition. This observation suggests that the ferroelectric ordering was not the only cooperative phenomenon involved below the transition temperature (156 K); there exists another order–disorder process, such as glass formation, at shorter length scales below 40 K. This second order–disorder process involves the freezing of the methyl-group internal rotations.

In summary, we have used ^1H NMR and specific heat measurements to study the motion of the central DMA⁺ moiety in DMAZF crystals. The moiety exhibits unique orientational dynamics with a decrease in the motional rate with temperature, resulting in an order–disorder phase transition at 156 K. Both the long-range and the short-range ordering behavior of this crystalline MOF were probed in detail and both have shown to be related to unique motional dynamics of the DMA⁺ moiety. At temperatures below 156 K, DMA⁺ starts to become more ordered with decreasing temperature. Initially this involves ordering of the amine NH_2 group. Below 40 K, the internal rotations of the CH_3 groups also start to freeze, leading to multiple states with closely spaced energies. This results in a memory effect in the temperature dependence of T_1^{-1} . Local coordination environments in the framework remain unchanged as no other anomalies are observed in the specific heat measurements. However, a peak is observed at about 25 K in a C_p/T^3 vs. T plot, which is a direct manifestation of a glass phase. The present study has thus revealed a previously undescribed feature of MOF compounds—the ability to exhibit glass-like properties, akin to that shown by many inorganic perovskites.

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